

## X-RAY INVESTIGATION OF DEPOSITS FORMED BY ELECTRIC ARC SPUTTERING COMPOSITES $Me_1-Me_2-C$

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### Introduction

In connection with the perspective of using nanostructural carbon in different fields of industry and techniques, the materials based on it attract heightened interest. Electronics gives a particular place to these materials due to their physical-chemical properties.

One of the products formed after the electric arc synthesis of fullerenes is deposit-a growth forming on the cathode. Deposit mainly consists of multi-wall carbon nanotubes and some quantity of graphitized mass of carbon.

As deposit is formed at high temperatures and under influence of power electric field, it in some or other form is of interest as the raw material for electronic industry.

Additions introduced into the deposit by intercalation, encapsulation and other methods have a strong influence on physical-chemical, electric and other properties of the product what permits the extension of their application.

### Results and discussion

In the work presented we have made X-Ray investigation of the products produced by electric arc evaporation of composites  $M_1+M_2+C$  (where  $M_1$  and  $M_2$  are different transition metals) We have used the following composites: (Hf+Ni+C); (Hf+Fe+C); (Co+Mg+C); (Co+Ti+C); (Co+W+C); (Co+Ni+C); (Fe+Ni+C); (Fe+Ni+Cr+C); (Si+C):

Previously the deposits were subjected to grinding during 3 hours in the planetary mill. X-Ray photography of the ground deposits was made on X-Ray apparatus DRON-3M in filter  $CuK\alpha$ -radiation followed by interpretation of the diffractograms. The position of the diffraction maximums was determined according to the center of gravity of lines.

X-Ray investigation of the deposit produced from the graphite anode without addition of metal catalysts showed that besides multi-layer carbon nanotubes the deposit consisted of two phase-hexagonal graphite and rhombohedral graphite with slightly increased interplane distances calculated by the center of gravity of the diffraction line (Fig.1).

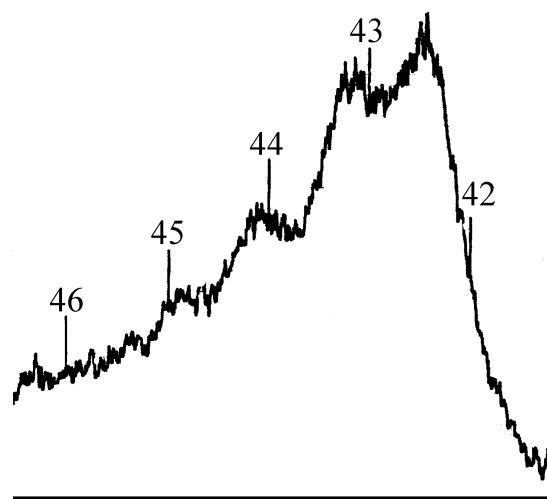


Fig. 1

After adding the metal catalysts the main phase in deposits remains graphite in two modifications (hexagonal phase predominates). Depending on the structure of the compositional addition, there appear carbides and intermetallides as well (Table 1).

So, after adding Si into the graphite row mix there appears silicon carbide  $\beta$  SiC (cubic modification with  $a=4,358\text{\AA}$ ) and traces of  $\alpha$ -SiC (hexagonal modification).

The composition (Hf+Ni) added into the graphite raw mix results in the formation of hafnium carbide HfC of the stoichiometric composition with the lattice constant  $a=4,636\text{\AA}$  and very small amount of  $Ni_3C$  (Fig. 2).

The composition (Hf+Fe) forms in the deposit the HfC phase of the stoichiometric composition and very small amount of solid solution Hf(Fe)C. The substitution of hafnium atoms for iron ones leads to the decreases in the lattice constant of hafnium carbide (Table 1)

Co+Mg added into the graphite row mix of the anode results in the formation on the cathode deposit magnesium carbide  $Mg_2C_3$  with the hexagonal lattice and very small amount of  $Co_2C$ .

Composite (Co+Ti) forms in the deposit titanium carbide with the stoichiometric composition.

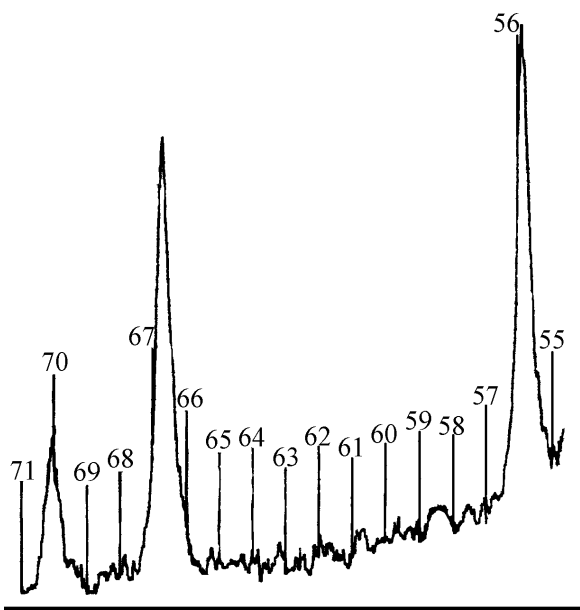


Fig. 2

Composite (Co+W) added creates in the deposit WC cubic and  $\alpha$ -WC hexagonal phases in insignificant amount.

(Co+Ni) added into the anode raw mix forms in the deposit continuous solid solution Ni (Co). The small increase in the nickel lattice constant indicates it. This increase is due to the larger Co atoms in comparison with NICKEL atoms (Table 1)

When composition (Fe-Ni) used, in the deposit solid solution  $\gamma$ -Fe (Ni) forms where Ni stabilizes the  $\gamma$ -Fe lattice. In deposit, after electric arc evaporation of the anode with addition of (Fe-Ni-Cr), we have observed  $\gamma$ -Fe (Ni) and  $Cr_3C_2$  phases (Table 1).

Table 1

Catalyst	Phase structure of deposit	$a$ , Å	$V$ , Å <sup>3</sup>	$R_{at}$ Me, Å
Si	C (graphite); $\beta$ SiC $\alpha$ -SiC	4,358	82,768	Si 1,17κ
Hf+Ni	C (graphite); HfC stoich. Ni <sub>3</sub> C traces	4,636	99,639	Hf 1,59
Hf+Fe	C (graphite); HfC stoich. Hf(Fe)C	4,636 4,605	99,639 97,653	Hf 1,59 Fe 1,27
Co+Mg	C (graphite); Mg <sub>2</sub> C <sub>3</sub> Co <sub>2</sub> C	–	–	Mg 1,60 Co 1,26
Co+Ti	C (graphite); TiC stoich. CoTi	4,324	80,846	Ti 1,45 Co 1,26
Co+W	C(graphite); WC cub. $\alpha$ -Wc hex.	4,220	75,151	W 1,40
Co+Ni	C(graphite); Ni(Co) s.sol.	3,529 3,524Ni ref.	43,950	Ni 1,24 Co 1,26
Fe+Ni	C(graphite); $\gamma$ -Fe(Ni) s.sol.	$\gamma$ -Fe(Ni) 3,595 $\gamma$ -Fe 3,645 ref.	46,462 48,427	Fe 1,27 Ni 1,24
Fe+Ni+Cr	C(graphite); $\gamma$ -Fe(Ni) sol. sol. Cr <sub>3</sub> C <sub>2</sub>	$\gamma$ -Fe(Ni) 3,598 $\gamma$ -Fe 3,645 ref.	46,578 48,427	Cr 1,28 Fe 1,27 Ni 1,24

### Conclusion

In the course of the work it has been established that the additions of transition metals in the evaporated electrodes during electric arc synthesis are presented in deposits as simple or complex carbides. Individual physical-chemical properties of each composition depend on the nature of the metal.

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